

Abstract

The strategy to design low stacking fault energy (SFE) eutectic $\text{CoCr}_{1.3}\text{FeNi}_{0.7}\text{MnNb}_x$ ($x = 0.3\text{--}0.45$) high entropy alloy (EHEA) and to enhance the properties through thermomechanical treatment have been systematically explored. EHEAs comprising of ultrafine lamellae of low SFE ($\text{SFE} = 11 \text{ mJ/m}^2$) face-centered cubic (FCC) and Fe_2Nb -type hexagonal-closed pack (HCP) Laves phase (C14-type) have been synthesized at cooling rates of $2.5\text{--}10 \text{ K/s}$ (arc melted ingots, AMIs) and $4.2 \times 10^2 \text{ K/s}$ (suction cast bars, SCBs). The effect of cooling rate on the microstructure evolution, their refinement and mechanical properties has been explored and are correlated with Nb addition, interlamellar size (λ_e), dislocation density (ρ_d), and twin density (ρ_t). Systematic investigation revealed that the phase interface strengthening, solid solution strengthening, and twin boundary strengthening play critical roles in providing the high strength (σ_y) up to 1.35 GPa along with 20% plastic strain (ε_f). The stress-strain curves and post-deformation analysis using transmission electron microscopy confirmed that the concurrent evolution of dislocations, deformation twins, and their mutual interactions, which are solely responsible for the large strain hardening ($n = 0.27$) at multiple stages during plastic deformation. The impact of thermo-mechanical processing on the microstructure evolution and mechanical properties of AMI EHEAs using cold forging and subsequent annealing have been investigated. The refinement of the microstructure, formation of dislocation substructure ($\rho_d = 1.5 \times 10^{15} \text{ m}^{-2}$), thermally stable deformation twins ($\rho_t = 7.8 \times 10^6 \text{ m}^{-1}$), and controlled precipitation of σ -phase ($5 \text{ vol.}\%$) reduce the mean free path (Λ) for dislocation glide and provide a good combination $\sigma_y = 1389\text{--}1537 \text{ MPa}$ and $\varepsilon_f = 11\%\text{--}13\%$. The stain rate jump test and the nanoindentation have been carried out at room temperature to investigate the effect of processing conditions on the rate sensitivity ($m = 0.0106\text{--}0.0111$) and activation volume ($V^* = 28b^3\text{--}41b^3$), which suggest that the deformation kinetics is mainly governed by the dislocation interactions with the ultrafine lamellae interface and the twin boundaries. The potential of warm-rolling to further tune the microstructure, and mechanical properties have been investigated. The present investigation further explore the effect of low SFE on the nucleation of twins/ or dislocations in the lamellar phase using various models. Theoretical calculations have been presented considering Stroh's pile-up model to establish the evolution of deformation bands and transfer of slip across the lamellae interface.

Keywords: Nano-/ultrafine eutectic microstructure, High entropy alloys, Stacking fault energy, Deformation twinning, Dislocation, Strengthening, Thermo-mechanical treatment.